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Recombination dynamics of localized excitons in a CdSe/ZnSe/ZnS\(_x\)Se\(_{1-x}\) single-quantum-well structure

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Optical properties of localized excitons have been studied in a highly strained CdSe quantum well with 1-ML thickness by employing time-resolved photoluminescence (PL) and nonlinear PL spectroscopy under various excitation conditions. At 20 K, the time-integrated PL from the well layer was peaked at 2.7276 eV with a linewidth of 21 meV under low excitation intensity (0.11 μJ/cm\(^2\)). The lifetime ranged from 200 ps to 50 ps as the monitored photon energy was changed from the low-energy tail to the high-energy one. The behavior could be well understood as a result of exciton localization, which is induced by terraces and islands with units of ML thickness fluctuation lying at the interface between CdSe and ZnSe. An emission with the fast decay component was observed at the low-energy side of the peak (2.7168 eV) under higher excitation condition. The emission could be well resolved as a positive component by the nonlinear PL measurement, and originates from the many-body effect of localized excitons, which is probably attributed to localized biexcitons. [S0163-1829(96)07828-9]

I. INTRODUCTION

Ternary and quaternary II-VI alloy semiconductors are examples of materials with perturbed crystal potential. The perturbation results from microscopic, statistical variations of the alloy composition on a microscopic scale. This alloy disorder leads to a density of tail states of the form of \(\exp[-E/E_0]^a\) extending down to the forbidden gap, where \(E\) is band energy and \(E_0\) the characteristic energy for the density of states. The value of \(n\) is under debate and it has been reported to be from 0.5 to 2.0. The localization of excitons to these tail states locates the peak energy of photoluminescence (PL) on the lower-energy side than that of absorption (the so-called Stokes shift) and the PL linewidth is broadened.

Such phenomena are also induced in quantum-well (QW) structures with interfacial disorders. Microscopically, heterointerfaces between wells and barriers are pseudosmooth within certain lateral regions, which vary in size at small terraces or extended islands with their step heights being usually in monolayers (ML). If the region of islands is substantially larger than the in-plane diameter of excitons, PL spectra are composed of a few sharp emissions, which correspond to quantized energy levels determined by the well width.\(^1,2\) Such an interface has been achieved in III-V semiconductors by the advanced growth technique. However, the density of islands highly depends on growth conditions and, in most cases, transition energies are averaged within the volume of excitons due to the small size of islands. Consequently, unlike the localization caused by the effect of alloy disorder, the degree of localization strongly depends on the quality of the interface.

In recent years, intense studies have been carried out on excitonic properties of ZnCdSe/Zn(S)Se QW’s in terms of their usefulness for blue-green laser diodes.\(^3,4\) Since the well layer in the structures is under compressive strain, the lowest excitonic transition occurs between the \(n=1\) conduction electron and \(n=1\) heavy hole, generally designated as \(E_{\text{lh}}\). The solid composition of Cd in Zn\(_x\)Cd\(_{1-x}\)Se well layers is typically about 15–20% in the device structures. The PL linewidth of \(E_{\text{lh}}\) for such layers grown under the appropriate condition is as low as about 9 meV at low temperature, a value that agrees fairly well with the theoretical one derived from the alloy broadening effect.\(^5,6\) This suggests that the localization of the exciton in these structures is mainly caused by the effect of alloy broadening.

In CdSe/ZnSe QW structures, on the other hand, localization of excitons occurs only due to interface roughness. This enables us to investigate easily the properties of localized excitons focusing on the interface structures. The well width fluctuation of 1 ML in the CdSe/ZnSe QW with ultrathin well width (\(\ll\) 5 ML) results in the fluctuation of the electron confinement energy of the order of 100 meV. The strong spatial variation of the carrier confinement leads to significant localization of excitons to the tail states, and the optical and electrical properties of the structure are extremely sensitive to such kind of disorder. Accordingly, another feature of this structure is the large lattice mismatch between the well and the barrier. It has been reported that pseudomorphic growth of the CdSe layer on the ZnSe buffer without the formation of misfit dislocations is limited to a few ML’s and the high PL efficiency is obtained in such pseudomorphic well layers.\(^7\) In this study, the CdSe single QW (SQW) with 1-ML thickness was embedded in a ZnS\(_x\)Se\(_{1-x}\)ZnSe/ZnS\(_x\)Se\(_{1-x}\) waveguide structure, which was coherently grown on the GaAs substrate. The use of the structure has two reasons: one is for the effective confinement of photogenerated carriers, and the other is that the flatness of ZnSe...
coherently grown on a ZnS$_x$Se$_{1-x}$ buffer is superior to that grown on a strain-relaxed ZnSe layer.

More recently there has been increasing interest in lasing processes in disordered semiconductors in terms of the fact that localized excitons can easily contribute to the optical gain due to their low density of states.\textsuperscript{3,8} In many studies, however, many-body effects of localized excitons have been ignored as a first-order approximation. This is probably because of the difficulty in the assignment of new emissions, which may appear as a function of photoexcitation intensity, caused by the inhomogeneous broadening effect such as alloying and/or interfacial disorder.

In this paper, effects of high excitation on the dynamical behavior of localized excitons in the CdSe SQW have been assessed by employing linear and nonlinear luminescence spectroscopy.

II. EXPERIMENTAL PROCEDURE

The CdSe/ZnSe/ZnS$_x$Se$_{1-x}$ separate confinement single quantum well (SC-SQW) structure described above was grown by molecular beam epitaxy on a Zn-doped $p$-type GaAs substrate oriented to (100). The growth of the SC-SQW structure proceeded as follows.

First, a 0.85-$\mu$m ZnS$_{0.075}$Se$_{0.93}$ cladding layer was deposited, followed by a 30-nm ZnSe buffer layer. A CdSe SQW (the active layer) was sandwiched between ZnSe waveguide layers (50 nm each). Finally, a 0.15-$\mu$m ZnS$_{0.075}$Se$_{0.93}$ capped layer was grown. The mean thickness of the well was set to 1 ML. The value was estimated according to the growth rate and PL energy of the CdSe layer. This estimation has been assured by the previous results for the growth of CdSe on ZnSe,\textsuperscript{9} where the CdSe thickness was estimated by the reflection high-energy electron diffraction oscillation and the cross section transmission electron microscopy.

Time-resolved PL measurements were carried out using a frequency-doubled beam of an Al$_2$O$_3$:Ti laser pumped by a cw Ar$^+$ laser, and a photon counting method with a synchronous streak camera in conjunction with a 25-cm single-grating monochromator. The wavelength (WL), the pulse width (PW), and the repetition rate (RR) of the excitation beam were 390 nm, 1.7 ps, and 82 MHz, respectively. The incident excitation energy density was tuned by the variable neutral density filter. The time resolution of the detection system is about 20 ps. Accordingly, the exponential decay time can be estimated within the accuracy of 5 ps by the deconvolution technique.

In order to investigate high excitation effects, nonlinear PL spectroscopy was performed by the population mixing technique, where two excitation beams were modulated at different frequencies ($\Omega$ = 831 Hz, $\Omega$ = 1000 Hz) by a chopper and focused at the same spot on a sample. The nonlinear PL component was obtained by detecting the sum-frequency element ($\Omega_1$ + $\Omega_2$ = 1831 Hz) of the total PL signal using a lock-in technique. In this study, the second harmonic light of the amplified laser output (WL, 400 nm: PW, 200 fs; RR, 250 kHz) was used as an excitation source, and the delay between the excitation pulse pairs was set to zero. The principle of this spectroscopy is based on the fact that the PL contains a substantial nonlinear component in the radiative process. For example, the negative nonlinearity is observed in the vicinity of the linear-PL peak under high excitation conditions due to the band-filling effect. Accordingly, positive nonlinearity is occasionally detected at either the higher- and/or lower-energy side of the negative peak as a result of the formation of hot excitons and/or bimolecular excitons (biexcitons).

III. RESULTS AND DISCUSSION

A. Time-resolved PL from a CdSe QW, ZnSe waveguides, and ZnS$_x$Se$_{1-x}$ cladding layers

Figure 1 shows a time-resolved PL spectrum from the CdSe/ZnSe/ZnS$_x$Se$_{1-x}$ SC-SQW structure under $I_{ex}$ = 0.11 $\mu$J/cm$^2$. Emissions from ZnSe and ZnS$_x$Se$_{1-x}$ layers decay rapidly, and the emission from the CdSe QW dominates in intensity. This indicates that photogenerated carriers are effectively captured to the well.

![PL spectrum](image)
hole is calculated to be 4.5 and 4.0 nm, respectively, which are values also achieved with the well width of about 4 nm. It should be also noted that the ZnSe waveguiding layers (~100 nm) act as the reservoir of photogenerated carriers, which promotes the collection of them to the lowest-energy level of the well. The FWHM of PL from the CdSe SQW is about 21 meV. The broadening originates from the fluctuation of well layer thickness within the volume of the exciton. This probably indicates that there exist some regions without the CdSe layer grown, and that the QW is divided into some parts in the lateral direction. If this is the case, it is necessary to take into account the effect of lateral confinement for the calculation of energy level.

B. Time-integrated PL

Figure 2 shows the time-integrated PL spectra at 20 K taken from the CdSe well layer under the $I_{ex}$ values of (i) 0.11, (ii) 0.93, and (iii) 4.72 $\mu$J/cm$^2$. The PL at the lowest excitation density shown in curve (i) is dominated by the recombination of localized excitons (denoted by $X_1$), whose peak is situated at 454.61 nm (2.7276 eV). The $X_1$ peak gradually shifts to higher photon energy with increasing excitation densities. This is because of the filling of the density state, which is easily induced for localized excitons lying at the tail state. This phenomenon is demonstrated by the nonlinear luminescence measurement described below.

With raising the excitation density up to 0.93 $\mu$J/cm$^2$, a shoulder appears on the low-energy side. When further increasing excitation density, the shoulder grows more rapidly than the $X_1$ line, and a new peak (denoted by $X_2$) appears at 456.41 nm (2.7168 eV). The energy difference between $X_1$ and the $X_2$ peaks is estimated to be about 11 meV.

C. Decay and lifetime

Figure 3 shows the lifetimes of the emission as a function of photon energy, together with the time-integrated PL spectrum measured under $I_{ex}=0.11$ $\mu$J/cm$^2$, the lowest excitation condition. Figure 4 depicts PL decay curves monitored at four different emission energies. The decay spectra are well fitted by single exponential curves. The PL intensity reaches the maximum after a certain time after pulsed excitation. The rise time as well as the lifetime increase with decreasing monitored photon energy. These behavior can be understood as the effect of exciton localization, where decay of excitons is not only due to radiative recombination but also due to transfer processes to the low-energy tail by the assistance of acoustic phonons.

Lifetimes obtained have been well fitted by the theoretical decay time $\tau(E)$ as a function of emission energy as follows:

$$\tau(E)^{-1} = \tau_r^{-1}[1 + \exp((E-E_{me})/E_0)],$$

where $\tau_r$ is the radiative lifetime, $E_{me}$ is the energy of the exciton, and $E_0$ is the energy separation between the exciton and the ground state.
where $E_0$ is the characteristic energy for the density of states, $\tau_r$ is the radiative lifetime, and $E_{me}$ is defined by a definite energy for which the decay time equals the transfer time.\(^\text{12}\)

The best fit has been obtained using the following parameters: $E_0=8.5$ meV, $\tau_r=209$ ps, and $E_{me}=2.7267$ eV. It should be noted that such a fitting under low excitation conditions has also been made for localized excitons in CdSe/ZnSe SQW’s grown by a self-limiting monolayer epitaxy\(^\text{7}\) and that a similar $E_0$ value has been obtained. $E_0$ represents the degree of localization. The value of 8.54 meV is larger than the values reported for CdS,$S_{1-x}$ (Refs. 12–21) and Zn$$_x$Cd$_{1-x}$S (Refs. 27 and 31) alloys and for Zn$_x$Cd$_{1-x}$Te/ZnTe (Refs. 25 and 26), and CdTe/ZnTe (Refs. 22 and 24) QW’s. This implies that the large localization of excitons is rather a universal characteristic in the ultrathin CdSe/ZnSe SQW system.

Figure 5 shows decay curves monitored at various emission energies under $I_{ex}=4.72$ $\mu$I/cm$^2$, the highest excitation condition. It was found that the PL transient $[I(t)]$ in the vicinity of the $X_2$ line decays with a double exponential profile shown as

$$I(t)=A_f \exp(-t/\tau_f) + A_s \exp(-t/\tau_s),$$

where $\tau_f$ and $\tau_s$ are the decay times of the fast and the slow components, respectively. Fitted values of $\tau_f$, $\tau_s$, and $A_f/A_s$ are plotted in Fig. 6 as a function of monitored emission energy. Both the slow and the fast decay components increase with decreasing photon energy, indicating that both the $X_1$ and the $X_2$ lines are related to localized excitons. The tendency of the slow components is almost the same as that of lifetimes obtained under $I_{ex}=0.11$ $\mu$I/cm$^2$. The fast components are estimated to be roughly half as small as the slow components.

The situation, however, is complicated in this case because the effect of transfer processes has to be considered in the localized excitonic states. Consequently, for the quantitative analyses, it is necessary to construct the model of population dynamics of localized excitons and localized biexcitons.

PL intensity, $I_{PL}$, changes with $I_{ex}$ as in the following equation. Neglecting the terms higher than third order, $I_{PL}$ is written as

$$I_{PL} = kI_{ex}^n aI_{ex} + bI_{ex}^2.$$ 

The signal detected by the nonlinear PL spectroscopy is the second term ($b$) proportional to $I_{ex}$.

Figure 7 shows the nonlinear PL spectra under three different excitation intensities. The positive signal means the superlinear component, while the negative signal corresponds to the sublinear one. Under low values of $I_{ex}$, the positive signal is observed in the whole spectral range, and the spectra shape is almost the same as the time-integrated linear PL. The origin is probably due to the saturation of residual nonradiative centers by the photogenerated carriers. Increasing $I_{ex}$ more than about 0.31 $\mu$I/cm$^2$, the positive signal at the $X_2$ line grows more than that at the $X_1$ line. Under the excitation intensity of $I_{ex}=4.72$ $\mu$I/cm$^2$, the nonlinear signal at the $X_1$ line changes its sign to show negative nonlinearity because of the band-filling effect of localized excitons. However, the positive signal at the $X_2$ line remains to grow even at the excitation density, indicating that the $X_2$ line originates from the many-body effect of localized excitons such as localized exciton-exciton collisions or localized biexcitons.\(^\text{34}\)

**D. Time evolution of PL spectra**

Figures 8 and 9 show the time evolution of the luminescence spectra measured as a function of time after pulsed excitation ($t_p$) under two different excitation intensities of 0.11 and 4.72 $\mu$I/cm$^2$, respectively. Under the excitation...
intensity of 0.11 \mu J/cm^2, the PL peak at \( t_d = 0 \) ps is located at 454.4 nm (2.728 eV), and then shifts towards a lower photon energy with delay time at a rate of 2.3 \times 10^7 \, \text{eV/s}.\) This behavior can be understood as a transfer process of excitons to the tail state as discussed in the previous section. The PL spectrum slightly narrows with time from \( t_d = 50 \) to 40 ps. A shoulder appears and grows at about 458 nm (~2.71 eV) after \( t_d \approx 40 \) ps. The component is overlapped with the main peak (\( X_1 \)) after \( t_d = 400 \) ps, and no peak shift was observed after then. This indicates that the density of tail states \( \rho(E) \) is not exactly the form of \( \exp[-E/E_0] \) but rather a form of separately distributed features depending on the distribution of islands and terraces in the QW.

Under \( I_{ex} = 4.72 \, \mu J/cm^2 \), the PL spectrum is nearly the same as that under \( I_{ex} = 0.11 \, \mu J/cm^2 \), and no \( X_2 \) line is observed just after the excitation (\( t_d = 0 \) ps). The \( X_2 \) line begins to grow at \( t_d \) equal to several tens of ps, and then decays more rapidly than the \( X_1 \) line. The dynamics of localized excitons under high excitation conditions can be described as follows. Photogenerated excitons relax to the states of lower energy and reach the local minima. In such a situation, excitons are localized spatially within the well layer due to the break of the momentum preservation, and many-body effects such as a formation of biexcitons or exciton-exciton scattering are easily produced compared to the case of flat QW’s without the effect of localization. It should be noted that a similar scenario has also been proposed by Kreller et al.\(^{36} \) in the Zn\(_{x}\)Cd\(_{1-x}\)Se/ZnSe system.

The energy difference between \( X_1 \) and \( X_2 \) is about 11 meV at \( t_d = 100 \) ps. The origin of the \( X_2 \) being due to inelastic collision of localized exciton, this energy difference should correspond to the binding energy of localized excitons. The exciton binding energy for bulk CdSe and ZnSe is 13 and 18 meV,\(^{37,38} \) respectively. We calculated the exciton binding energy \( E_{ex} \) of the CdSe/ZnSe QW’s by the variational method using the trial wave function of the 1S exciton as

\[
\psi_{1S} = \exp\left(-\frac{1}{\lambda} \sqrt{x^2 + y^2 + \eta(z-z_h)^2}\right).
\]

The result shows that the maximum \( E_{ex} \) (about 34 meV) is achieved at the well width \( (L_w) \) of 3 ML, and that \( E_{ex} \) is estimated to be 30.5 meV at \( L_w = 1 \) ML. Although analyzing the \( E_{ex} \) value in the strongly localized system is diff-
ficult, it is expected that the $E_{\text{ex}}$ value will be increased compared to the case of no localization effect, due to the enhanced confinement of wave functions. Thus the process of inelastic collision can be excluded. It is very difficult to estimate theoretically the biexciton binding energy in the system where the penetration of wave functions to the barrier has to be taken into account. Nevertheless, judging from the relationship of decay times between the $X_1$ and $X_2$, we could conclude that the radiative recombination of localized biexcitons is the most probable process for the $X_2$.

IV. CONCLUSIONS

Optical properties of localized excitons in a highly strained CdSe QW with about 1-ML thickness have been studied at 20 K by employing time-resolved PL and nonlinear PL spectroscopy under various excitation conditions. The time-integrated PL from the well layer was peaked at 2.7276 eV with the linewidth of 21 meV under low excitation intensity (0.11 $\mu$J/cm$^2$). The lifetime ranged from 200 to 50 ps as the monitored photon energy was changed from the low-energy tail to the high-energy one. These features could be interpreted by the model of exciton localization, which is induced by terraces and islands lying at the interface between CdSe and ZnSe layers.

An emission with the fast decay component was observed at the low energy side of the peak (2.7168 meV) under higher excitation condition. The component was also detected as a superlinear signal in nonlinear PL spectra, and was attributed to a dense excitonic effect such as localized biexcitons.

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11 It should be noted that this is a rough estimation because the concept of the carrier envelope function is valid only for relatively thick layers, and, for a more rigorous way, the effective-mass theory should be replaced by a more correct tight-binding method in the case of ultrathin well thickness.


