# Pentadiamond: A Hard Carbon Allotrope of a Pentagonal Network of $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3} \mathrm{C}$ Atoms 

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#### Abstract

A pentagonal covalent network consisting of $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3} \mathrm{C}$ atoms has been investigated based on the density functional theory. Our theoretical investigations clarified that the pentagonal covalent network is a metastable three dimensional carbon allotrope with the $F m \overline{3} m$ space group possessing remarkable mechanical properties: Relatively high bulk modulus of 381 GPa together with a negative Poisson's ratio of -0.241 . Accordingly, the pentagonal covalent network possesses extremely high Young's and shear moduli of 1691 and 1113 GPa , respectively, surpassing those of the diamond. The electronic structure of the pentagonal network is a semiconductor with an indirect band gap of 2.52 eV between L and X points for valence and conduction band edges, respectively, with the relatively small carrier masses.


Carbon is a unique element in terms of the structural variation in its possible allotropes [1], owing to the three possible orbital hybridizations, boundary conditions, and topological defects. Three forms of orbital hybridization, $\mathrm{sp}, \mathrm{sp}^{2}$, and $\mathrm{sp}^{3}$, render allotropes covering all dimensions: C with the $\mathrm{sp}, \mathrm{sp}^{2}$, and $\mathrm{sp}^{3}$ hybridization forms one-dimensional C chains (polyyne)[2, 3], two-dimensional sheet (graphene)[4], and three dimensional network (diamond), respectively, which are well known conventional carbon allotropes. In addition, the boundary conditions and topological defects cause further allotropes with unique morphologies, such as fullerene [5] and carbon nanotubes [6], which possess unusual electronic and structural properties depending on both local and global network topologies [7-14]. For instance, fullerenes basically possess spherical harmonic $Y_{l m}(\theta, \phi)$ electron states with the deep lowest unoccupied states owing to their hollow cage network topology caused by 12 pentagonal rings embedded in appropriate number of hexagons [7-11]. Moreover, networks consisting of pentagonal rings of $\mathrm{sp}^{2}$ or $\mathrm{sp}^{3} \mathrm{C}$ atoms causes unusual physical properties, such as magnetism on two dimensional $\mathrm{sp}^{2} \mathrm{C}$ networks [15] and negative Poisson's ratio on pentagraphene [16, 17].

Carbon allotropes consisting of both $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3} \mathrm{C}$ atoms have been attracting much attention, because of their morphological diversity caused by the huge number of combinations and of arrangements of $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3} \mathrm{C}$ atoms in the networks. High-pressure and hightemperature treated $\mathrm{C}_{60}$ and higher fullerenes are representative examples of such allotropes, being expected to possess low mass density owing to their covalent bonds between fullerenes and nanoscale pore arising from fullerene cages [18-21]. The rapid quenching of amorphous carbon causes a $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3}$ hybrid C allotrope, known as the Q-carbon, which has been reported to possess magnetism and remarkable hardness surpassing diamond [22, 23]. In addition to these all carbon materials, the controlled polymerization of hydrocarbon molecules causes two- and three-dimensional covalent hydrocarbon networks of $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3} \mathrm{C}$ atoms by assembling appropriate linker and linkage hydrocarbon molecules [24-28]. Such bottom-up synthesis techniques ensure that the novel carbon allotropes with unusual physical properties are expected by assembling and polymerizing appropriate hydrocarbon molecules $[29,30]$. These morphologically and structurally well-defined covalent hydrocarbon networks are considered to be applicable for wide area of modern technologies, such as gas storage, catalytic application, and optoelectronics, because of their porous structure with stable covalent framework and unusual electronic structures.

In this work, we aim to theoretically explore a three-dimensional carbon allotrope consisting of $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3} \mathrm{C}$ atoms (pentadiamond), owing to co-polymerization of hydrocarbon molecules containing pentagonal rings (spiro[4.4]nona-2,7-diene [31, 32] and [5.5.5.5]fenestratetraene [33-36]) as for a possible candidate of hard carbon allotropes with unusual mechanical properties, using the density functional theory with the generalized gradient approximation. The pentadiamond sorely consists of pentagonal rings with the $F m \overline{3} m$ space group owing to the copolymerization of these constituent molecules. Our calculations demonstrated that the pentadiamond has the high bulk modulus of 381 GPa which is approximately $80 \%$ of that of diamond, indicating that the pentadiamond is the hard carbon allotrope. Furthermore, it has negative Poisson's ratio of -0.241 , leading to extremely high Young's and shear moduli of 1691 and 1113 GPa , respectively, which is higher than those of diamond and other ultrahard carbons. The pentadiamond is a semiconductor with an indirect band gap of 2.25 eV expected to possess high hole mobility.

All calculations were conducted on the basis of the density functional theory (DFT) [37, 38] implemented in the program package STATE [39-41]. We used the generalized gradient approximation (GGA) with the Perdew-Burke-Ernzerhof functional to describe the exchange-correlation potential energy among interacting electrons [42]. A ultrasoft pseudopotential generated using the Vanderbilt scheme is adopted for describing the interaction between electrons and ions [43]. Valence wave functions and deficit charge densities were expanded in terms of plane-wave basis sets with cutoff energies of 25 and 225 Ry, respectively, which give sufficient convergence in the total energy and electronic structures of covalent networks consisting of both $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3} \mathrm{C}$ atoms [44]. Ab initio molecular dynamics simulations were conducted using the velocity scaling method to keep the temperature constant during the simulation to insure the thermal stability of the pentadiamond. Integration over the first Brillouin zone was carried out using equidistant $4 \times 4 \times 4$ and $8 \times 8 \times 8 \mathrm{k}$-meshes for the self-consistent electronic structure calculations and for the density of states calculations, respectively. Lattice parameters and internal atomic coordinates were fully optimized until the force acting on each atom became less than $5 \mathrm{mRy} / \AA$.

Figure 1 shows an optimized geometry of the pentadiamond with the lattice parameter of $9.195 \AA$ and the $F m \overline{3} m$ space group. The covalent network only consists of pentagons in which three of five edges are shared by its adjacent pentagons [Fig. 1(a)], owing to the copolymerization of spiro[4.4]nona-2,7-diene and [5.5.5.5]fenestratetraene which are alter-
(a)

(b)


FIG. 1. (a) An optimized geometry of the pentadiamond under an optimum lattice constant of $9.195 \AA$ and the $F m \overline{3} m$ space group. Pale and dark gray balls indicate $\mathrm{sp}^{3}$ and $\mathrm{sp}^{2}$ carbon atoms, respectively. Indexes $\mathrm{C} 1, \mathrm{C} 2$, and C 3 indicate the independent atomic site under the $F m \overline{3} m$ space group. (b) Spiro[4.4]nona-2,7-diene and [5.5.5.5]fenestratetraene are alternately arranged at the vertexes of the cubic lattice. White and gray balls indicate H and C atoms. respectively.
nately arranged in the vertexes of the cubic lattice [Fig. 1(b)]. According to these constituent molecules and their arrangement, a unit cell of the petadiamond contains 22 C atoms, 10 and 12 of which are $\mathrm{sp}^{3}$ and $\mathrm{sp}^{2} \mathrm{C}$ atoms, respectively. In addition, the symmetry operators belonging to the $F m \overline{3} m$ group reduce the independent atomic sites to three, whose atomic coordinates are listed in Table I. Because the network consists of $\mathrm{sp}^{2}$ (threefold coordinated) and $\mathrm{sp}^{3}$ (fourfold coordinated) C atoms, covalent bonds are classified into two groups. Cal-

TABLE I. Fractional atomic coordinates of the pentadiamond with the $F m \overline{3} m$ space group under the optimized lattice parameter of $9.195 \AA$. Atomic indexes are defined in Fig. 1(a).

| C1 | 0.250 | 0.250 | 0.250 |
| :--- | :--- | :--- | :--- |
| C2 | 0.152 | 0.152 | 0.152 |
| C3 | 0.198 | 0.198 | 0.000 |

culated bond lengths associated with the $\mathrm{sp}^{3} \mathrm{C}$ atom are 1.563 and $1.520 \AA$ for $\mathrm{C} 1-\mathrm{C} 1$ and C2-C3 bonds, respectively, reflecting their $\mathrm{sp}^{3}$ bond nature. In contrast, the bond length associated with $\mathrm{sp}^{2} \mathrm{C}$ atoms (C3-C3 bond) is $1.349 \AA$, indicating their double bond nature of $\mathrm{sp}^{2} \mathrm{C}$ atoms. As shown in Fig. 1(a), the pentadiamond possess large cubic pores with $3.644 \AA$ edges surrounded by the pentagonal covalent network. Therefore, it has a low mass density with the density of $2.26 \mathrm{~g} / \mathrm{cm}^{3}$, which is the same as that of graphene while smaller by $36 \%$ than that of diamond.

The pentadiamond has relative total energy of 275 meV /atom with respect to that of diamond (Fig. S1). The total energy is higher than that of diamond, graphite, and other hard carbon materials (Table II), but still lower than that of $\mathrm{C}_{60}$ which is known to be metastable zero-dimensional carbon allotrope. The moderate total energy is ascribed to the structural distortion of bond angles for both $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3} \mathrm{C}$ atoms. For the $\mathrm{sp}^{3} \mathrm{C}$ atoms, although the C 1 atom has almost perfect $\mathrm{sp}^{3}$ hybridization with the bond angle of $\theta_{212}=109.4^{\circ}$, the C 2 atom has bond angles of $\theta_{123}=115.9^{\circ}$ and $\theta_{323}=101.9^{\circ}$, which are wider and narrower than the bond angle of an ideal $\mathrm{sp}^{3}$. For the $\mathrm{sp}^{2} \mathrm{C}$ atoms, because of the pentagonal network, the bond angle associated with the C 3 is $\theta_{232}=133.4^{\circ}$ and $\theta_{233}=113.3^{\circ}$ which are significantly larger and smaller, respectively, than the $\mathrm{sp}^{2}$ bond angle. The moderate energy also increases the formation energy $\Delta E$ of the pentadiamond on the direct copolymerization of spiro[4.4]nona-2,7-diene and [5.5.5.5]fenestratetraene:

$$
\text { spirononadiene }+ \text { fenestratetraene }=\text { pentadiamond }+12 \mathrm{H}_{2}+\Delta E
$$

The calculated formation energy is $0.31 \mathrm{eV} /$ atom, reflecting the energy cost to form covalent network of $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3} \mathrm{C}$ with distorted bond angles. Thus, the pentadiamond is expected to be synthesized using the Yamamoto-type Ullmann cross coupling [26] on bromospiro[4.4]nona-2,7-diene and bromo[5.5.5.5]fenestratetraene instead of their pristine


FIG. 2. Phonon dispersion and phonon density of states of pentadiamond.
form as the cases of polymerization reactions of hydrocarbons. Furthermore, the selforganization process of the precursors with appropriate functional groups in liquid is the other possible procedure as the case of covalent organic frameworks [24].

We investigate the thermal stability of the pentadiamond by conducting ab initio molecular dynamics (MD) simulations under the temperature of 4000 K . To explore the possibility of long-range structural distortions, we perform the constant-temperature MD calculations on the extended cell, which contains 88 C atoms, and the primitive $1 \times 1 \times 1$ cell for 14 and 146 psec simulation times, respectively. After the 14 psec MD simulation on the pentadiamond with the extended cell at the temperature of 4000 K , the pentadiamond keeps its initial covalent network topologies. As for the primitive cell simulation, the pentadiamond also retain their covalent network up to 146 psec simulation time. In addition, there are

TABLE II. Total energy, bulk modulus, Poisson's ratio, averaged Young's modulus, and averaged shear modulus of the pentadiamond and diamond. The total energy and mechanical constants of the other hard and ultrahard carbon mateirals, Q-carbon, M-carbon, W-carbon, and BCT C4, in literatures are also listed. The LDA in parentheses indicates the energy evaluated using the local density approximation with the Perdew-Zunger functional [50].

|  | Energy | Bulk modulus Poisson's ratio Young's modulus |  | Shear modulus |  |
| :---: | :---: | :---: | :---: | :---: | :---: |
|  | $(\mathrm{meV} /$ atom $)$ | $(\mathrm{GPa})$ |  | $(\mathrm{GPa})$ | $(\mathrm{GPa})$ |
| Pentadiamond | 275 | 381 | -0.241 | 1691 | 1113 |
|  | $383(\mathrm{LDA})$ | - | - | - | - |
| Diamond (theory) | - | 468 | 0.046 | 1273 | 608 |
| Diamond(exp.) | - | $433[45]$ | $0.069[46]$ | $1143[46]$ | $502[45]$ |
| Q-carbon [47] | - | 616 | 0.084 | 1538 | 709 |
| M-carbon [48] | 171 | 400 | - | - | 406 |
| W-carbon [48] | 151 | 403 | - | - | 451 |
| BCT C4 [49] | 193 (LDA) | 421 | - | 973 | 436 |

no soft vibration modes in the phonon spectra, indicating that the pentadiamond is dynamically stable (Fig. 2). The phonon dispersion calculation was performed with density functional perturbation theory [51] implemented in the QUANTUM ESPRESSO code [52]. Therefore, the pentadiamond is thermally and energetically stable, once it was synthesized using appropriate experimental schemes.

The pentadiamond possesses remarkable mechanical properties summarized in Table II. The mechanical properties are studied using the elastic constants (stiffness) $c_{i j}$, which is evaluated by taking the finite difference of the total energy with respect to strains. Calculated elastic constants are 1715.3, -283.5 , and 1187.5 GPa for $c_{11}\left(=c_{22}=c_{33}\right), c_{12}\left(=c_{13}=c_{23}\right)$, and $c_{44}\left(=c_{55}=c_{66}\right)$, respectively. Note that the elastic constants obviously satisfy the Born stability criterion ( $c_{11}-c_{12}>0, c_{11}+2 c_{12}>0$, and $c_{44}>0$ ), corroborating the structural stability of the pentadiamond. Under the cubic symmetry, the bulk modulus $B$ is calculated using the form

$$
B=\frac{c_{11}+2 c_{12}}{3}
$$



FIG. 3. (a) Calculated Young's modulus, (b) shear modulus, and (c) Poisson's ratio of the pentadiamond as a function of the lattice orientations. Solid and dotted lines indicate the positive and negative values, respectively.

The calculated bulk modulus $B$ is 381 GPa which is over $80 \%$ of that of diamond and close to those of the other hard and ultrahard carbon materials (Table II), indicating the pentadiamond is a potential candidate of hard carbon allotropes, even though it has low density as that of graphite.

To further investigate the mechanical properties, we calculate the Young's modulus of the pentadiamond using the equation,

$$
E=\frac{1}{s_{11}-2\left(s_{11}-s_{12}-\frac{s_{44}}{2}\right)\left(\cos ^{2} \theta \sin ^{2} \theta+\sin ^{2} \theta \cos ^{2} \phi \sin ^{2} \phi\right)}
$$

where $\theta$ and $\phi$ are the Euler's angles and $s_{i j}$ are the elastic compliance evaluated by $c_{i j}$ with relations of $s_{11}=\frac{c_{11}+c_{12}}{\left(c_{11}-c_{12}\right)\left(c_{11}+2 c_{12}\right)}, s_{12}=\frac{-c_{12}}{\left(c_{11}-c_{12}\right)\left(c_{11}+2 c_{12}\right)}$, and $s_{44}=\frac{1}{c_{44}}$. The calculated Young's modulus is shown in Fig. 3(a), indicating that the pentadiamond has extremely high Young's modulus over 1.5 TPa for all directions. Using the Young's and bulk moduli, we further evaluated the shear modulus [Fig. 3(b)]. As the Young's modulus, the pentadiamond also possesses the high shear modulus over 1 TPa for all directions. Therefore, the pentadiamond may exhibit extreme stiffness against anisotropic structural deformations. Note that the Young's and shear moduli are higher than those of other hard and ultrahard carbon allotropes (Table II). These large moduli imply that the pentadiamond possesses the negative Poisson's ratio. Indeed, the calculated Poisson's ratio exhibits negative values
ranging from -0.20 to -0.28 , depending on the lattice directions [Fig. 3(c)]. Furthermore, the remarkable mechanical properties causes the high sound speed of $28700 \mathrm{~m} / \mathrm{s}$, which is higher than those of the diamond and other bulk carbon materials. Because the sound speed is expressed as $v_{s}=\sqrt{\frac{3 B}{\rho} \frac{1-\nu}{1+\nu}}$ where $B, \rho$, and $\nu$ are the Bulk modulus, density, and Poisson's ratio, respectively, the remarkable sound speed is ascribed to the negative Poisson's ratio. Accordingly, the pentadiamond is a hard carbon allotrope with unusual mechanical response on the tensile and compressive strain, being expected to constituent material for wide area of modern technologies.

Figure 4 shows the electronic structure and density of states (DOS) of the pentadiamond. The pentadiamond is a semiconductor with an indirect band gap of 2.52 eV . The valence band top and the conduction band bottom are located at the L and X points, respectively. The highest branch of the valence band and the lowest branch of the conduction band possess the substantial dispersion of 1 eV or more. Therefore, the pentadiamond is expected to have small effective masses around its band edges: Calculated electron masses at the X are 0.98 and $0.67 m_{e}$ along directions to the $\Gamma$ and W points, respectively, which are lighter than that of the diamond. In contrast, as for the valence band edge, calculated hole masses at the L point are 1.59 and $0.76 m_{e}$ along directions to the W and $\Gamma$ points, respectively, being heavier than that of the diamond. The moderate carrier mass and the large density of states at the band edges imply that the pentadiamond may possess moderate carrier mobility for both electron and hole. The dispersion relation as well as the density of state show that the pentadiamond possesses three-dimensional bulk electronic structure as that of diamond, reflecting its three dimensional covalent network with the high symmetry.

To provide further insight into the electronic structure of the pentadiamond, we investigate the squared wave function of the highest branch and the lowest branch of the pentadiamond at particular symmetry points (Fig. 5). The wave functions of the highest branch of the valence band at the L and $\Gamma$ points are distributed on the C 3 atoms with bonding $\pi$ state nature, owing to their $\mathrm{sp}^{2}$ hybridization. The wave function of the lowest branch of the conduction band at the $\Gamma$ and X points is also distributed on C 3 atoms with antibonding $\pi$ nature. Note that both the valence and conduction states are not pure $\pi$ state but hybridized states containing small amount of $\sigma$ component. These facts imply that the electron states near and around the band edges are regarded as the $\pi$ electron states of $\mathrm{sp}^{2} \mathrm{C}$ dimers which are separated by about $2.6 \AA$ from its 8 adjacents. The large band dispersion is ascribed
to the substantial overlap of $\pi$ states between adjacent dimers of C 3 atoms thanks to the relatively small spacing between them compared with that of the conventional molecular solids with $\pi$ electrons.

We theoretically explore the possibility of the three-dimensional allotrope of $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3}$ C atoms by assembling adequate hydrocarbon molecules as the constituents, using DFT with GGA. We found that the copolymerization of spiro[4.4]nona-2,7-diene and [5.5.5.5]fenestratetraene with the $F m \overline{3} m$ symmetry causes the three-dimensional covalent pentagonal network of $\mathrm{sp}^{2}$ and $\mathrm{sp}^{3} \mathrm{C}$ atoms named the pentadiamond as the metastable carbon allotrope. The pentadiamond has the moderate total energy of 275 meV with respect to that of diamond and remarkable thermal stability up to 4000 K . According to covalent network of pentagonal ring, the pentadiamond are possible candidate of the hard carbon material with the remarkable mechanical properties: The bulk modulus is close to that of the diamond, and the Young's and shear moduli are higher than these of diamond, although it has remarkably low density which is the same as graphite. Furthermore, the pentadiamond has negative Poisson's ratio of -0.24 . The pentadiamond is a semiconductor with an indirect band gap of 2.52 eV between the L and X points for the valence and conduction band edges, respectively, where the moderate carrier mobility is expected by the carrier effective masses and the DOS at the band edges. The detailed wave function analysis unraveled that the electron states near the band edges are ascribed to the $\pi$ electrons distributed on $\operatorname{sp}^{2} \mathrm{C}$ atoms.

The author thanks JST-CREST Grant Numbers JPMJCR1532 and JPMJCR1715 from the Japan Science and Technology Agency, JSPS KAKENHI Grant Numbers JP20K05253, JP20H02080, JP20H00316, JP17H01069, and JP16H06331 from the Japan Society for the Promotion of Science, the Joint Research Program on Zero-Emission Energy Research, Institute of Advanced Energy, Kyoto University, and University of Tsukuba Basic Reseach Support Program (S). Part of the calculations was performed on an NEC SX-Ace at the Cybermedia Center at Osaka University.

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FIG. 4. The electronic band structure and density of states of the pentadiamond along the high symmetry points and axes. The energies are measured from that of the valence band top.


FIG. 5. Isosurfaces of the squared wave function of the pentadiamond of the highest branch of the valence band at (a) $\Gamma$ and (b) L points, and those of the lowest branch of the conduction band at (c) $\Gamma$ and (d) X points. Pale and dark gray balls indicate $\mathrm{sp}^{3}$ and $\mathrm{sp}^{2}$ carbon atoms, respectively. The isosurfaces correspond with the electron density of $3.5 \mathrm{e} /(\mathrm{a} . \mathrm{u} .)^{3}$.


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