観察と分析：ネイターンによるシングルイベントオーバーバーナー

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Abstract— Annular microvoids formed by neutron-induced single-event burnout (SEB) in Si power diodes were observed by a slice-and-view technique. The axial symmetry of damage region reflects the spatially-isotropic thermal diffusion that occurred. Analytical formulae for the local rise in temperature during SEB were derived from the thermal diffusion equation. The local temperature was found to increase in direct proportion to the deposited energy, which was expressed as the time integration of the product of the applied voltage and the SEB current. This current is the result of charges generated by recoil ions and subsequent current-induced avalanche. The diameter of the damage region was estimated using the analytical formulae and the energy associated with Joule heating, which was calculated by TCAD device simulations, and was found to be comparable in size to the observed annular voids. The SEB current density was also calculated based on the simulated SEB current and the size of the damage region.

Index Terms— Single event burnout (SEB), white-neutron irradiation, annular microvoids, thermal diffusion equation, current-induced avalanche (CIA)

I. INTRODUCTION

HIGH ENERGY particles at sea level are produced by nuclear spallation reactions between cosmic rays and atmospheric nuclei. There is general agreement that 97% of these particles are white neutrons with a broad energy distribution [1]. Cosmic-ray-induced neutrons cause catastrophic failure in power devices in a stochastic manner [2]-[6]. The failure rate associated with single-event burnout (SEB) increases sharply with applied voltage when the voltage exceeds a certain threshold value. It is therefore crucial to optimize the SEB threshold voltage during the device design stage in order to achieve highly reliable devices and avoid chance failures.

There have been several numerical investigations of the SEB failure mechanism [7]-[15]. In addition, white-neutron irradiation experiments and transient thermal simulations have been performed to analyze the SEB triggering mechanism in silicon (Si) and silicon carbide (SiC) power devices [16]-[25]. A nuclear reaction between individual incident neutrons and Si nuclei induces recoil ion formation, and electron-hole pairs are generated along the ion track. A highly localized current then flows through the power device, and local thermal destruction can occur if the voltage applied to the device is higher than the SEB threshold voltage.

Transient thermal simulations have been used to investigate the behavior of electron-hole pairs generated along the ion tracks in power devices under the influence of the applied voltage. The electrostatic potential adopts a funnel-like shape and the highly localized current results in an increase in the electron density in the vicinity of the n' drift region/n” diffusion region interface [16], [18]-[20], [22]. This space charge effect leads to a shift of the peak electric field from the p’ body region/n’ drift region interface to the n’ drift region/n” diffusion region interface. Consequently, the electric field distribution assumes a hammock-like shape along the ion track, which corresponds to a region of negative differential resistance (NDR) [26], [27]. Punch-through at the anode contact subsequently occurs with increasing current density, and this corresponds to a region of positive differential resistance (PDR) [26]. Therefore, the diode behaves locally like a resistor. It was clarified that the SEB mechanism in power diodes is thermal failure caused by local secondary breakdown. Moreover, it was shown to be possible to control the SEB threshold voltage by optimal choice of device parameters such as the drift region thickness [16]-[20].

In the present study, annular microvoids formed by neutron-induced SEB in Si power diodes were observed using a slice-and-view technique. In addition, analytical formulae for the local rise in temperature during SEB were derived from the thermal diffusion equation. The diameter of the damage region was then estimated using these formulae together with the energy associated with Joule heating, which was determined by technology computer aided design (TCAD) device simulations. The SEB current density was also estimated based on the simulated SEB current and the size of the damage region.

II. ANALYSIS OF NEUTRON-INDUCED SEB USING A SLICE-AND-VIEW PROCEDURE

In the present study, white-neutron irradiation of Si power diodes was carried out. A schematic diagram of the experimental test circuit is shown in Fig. 1. A voltage of 1050 V, which is close to the SEB threshold, was applied between the cathode and anode during neutron irradiation. Fig. 2 shows the
chip failure rate as a function of the applied voltage for Si power devices. The failure rate is seen to increase sharply at a particular voltage, which is defined as the SEB threshold voltage [20]. Transient device simulations were carried out to clarify the SEB triggering mechanism for Si diodes, and the results are shown in Fig. 3. The applied voltage was 1050 V, and SEB was found to occur at this voltage. It can be seen that the aluminum surface remains at room temperature during the SEB event.

Fig. 4 shows a scanning electron microscopy (SEM) image of the photoemission region at the anode identified using backside photoemission microscopy. The surface of the Al electrode showed no damage due to SEB. Fig. 5 shows SEM images of the Si surface after Al etching. A crater with an elevated rim is seen to be present, which is thought to be associated with motion of molten Si from within the device to the surface. Fig. 6(a) shows slice-and-view observations using FIB. The SEM image is at 0.4 μm depth from the Si surface. Two damage regions with a spacing of 2 μm can be observed, one of these regions is indicated by a yellow rectangle. The p+ diffusion layers in the anode region were indicated by the blue rectangles as shown in Fig. 6(b). Transient thermal device simulations were carried out for the cross section indicated by the dashed red line connecting the two damage regions in Fig. 6(b). As shown in Fig. 6(d), hole current crowding occurs at the corners of the p+ diffusion layers, with a spacing of 2 μm. Consequently, the lattice temperature rises at the two points shown in Fig. 6(e). Therefore, the two damage regions indicate traces that branched hole currents flowed into the two p+ diffusion patterns. Fig. 7 shows slice-and-view SEM images of the damage region indicated by the yellow rectangle in Fig. 6(a). A unique annular microvoid with a diameter of about 1 μm was observed in the p+ body region on the anode side, as shown in Figs. 7(a)-(c). The presence of this void suggests that the maximum lattice temperature in the device reached the melting point of Si due to the highly localized current caused by SEB. When this occurred, atoms in the melted region migrated to the device surface owing to thermal expansion of the Si. An annular void was then formed at the circumference of the damage region during the cooling process, and Si erupted above the device surface, as shown in Fig. 5(b). The axisymmetric shape of the void reflects the spatially isotropic thermal diffusion that occurred.

As shown in Fig. 7(d), the void was no longer observed at a depth of 1.4 μm from the Si surface (inside the p+ body region on the anode side). At a depth of 2.8 μm, a relatively straight crack was observed, as shown in Fig. 7(e). Furthermore, linear cracks were formed on both sides of the melt-solidification region located in the center of the damage region, as shown in Fig. 7(f). The cracks gradually began to disappear in deeper regions, as seen in Figs. 7(g)-(o), and were completely gone at a depth of 9.8 μm, as shown in Fig. 7(p). Typically, cracks in materials originate at edges and surfaces, and then propagate inwards. However, the cracks due to SEB are unique in shape and only form inside the device.

Fig. 8 shows an analysis of the annular void region using high-angle dark-field scanning transmission electron microscopy (HAADF-STEM) and energy dispersive X-ray spectroscopy (EDX). A voltage of 1023 V, which is close to the SEB threshold, was applied to the sample during white-neutron irradiation, so that the chip failure rate due to SEB was a function of the applied voltage, as shown in Fig. 2.
Fig. 4. SEM images of emission spot at anode identified using backside photoemission microscopy: (a) Low-magnification image and (b) high-magnification image.

Fig. 5. SEM images of Si surface after aluminum etching: (a) Low-magnification image, and (b) high-magnification image.

Fig. 6. Slice-and-view observations using FIB: SEM image at 0.4 μm depth from the Si surface (a). Device simulation results at the broken red line shown in (b) when the maximum lattice temperature reaches the Si melting point: (c) doping profile (1/cm²), (d) hole current density (A/cm²) and (e) lattice temperature (K).

Fig. 7. Slice-and-view SEM images of damage region indicated by yellow rectangle in Fig. 6 (a):
(a) depth of 0.8 μm from the Si surface (inside p+ body on the anode side)
(b) depth of 1.0 μm from the Si surface (inside p+ body on the anode side)
(c) depth of 1.2 μm from the Si surface (inside p+ body on the anode side)
(d) depth of 1.4 μm from the Si surface (inside p+ body on the anode side)
(e) depth of 2.8 μm from the Si surface (inside n+ drift region)
(f) depth of 3.0 μm from the Si surface (inside n+ drift region)
(g) depth of 3.2 μm from the Si surface (inside n+ drift region)
(h) depth of 4.0 μm from the Si surface (inside n+ drift region)
(i) depth of 4.5 μm from the Si surface (inside n+ drift region)
(j) depth of 5.3 μm from the Si surface (inside n+ drift region)
(k) depth of 5.6 μm from the Si surface (inside n+ drift region)
(l) depth of 6.3 μm from the Si surface (inside n+ drift region)
(m) depth of 7.2 μm from the Si surface (inside n+ drift region)
(n) depth of 7.8 μm from the Si surface (inside n+ drift region)
(o) depth of 8.8 μm from the Si surface (inside n+ drift region)
(p) depth of 9.8 μm from the Si surface (inside n+ drift region)
The void shown in Fig. 8 has a diameter of 0.34 μm, which is smaller than the value of 1 μm for the void shown in Fig. 7. Thus, the extent of the SEB damage region is variable. In the HAADF-STEM image, there is a contrast difference between the inside and outside of the annular void. This reflects the melt-solidification that occurred within the damage region due to the highly localized SEB current. Moreover, the Al elemental map shows that Al atoms have migrated from the anode to the highly localized SEB current. Moreover, the Al elemental map shows that Al atoms have migrated from the anode to the inside of the device.

This can be approximated as

\[
\rho \tau E V \equiv \int_0^{t_f} I(\tau) \cdot V(\tau) \, d\tau.
\]

where \( t_f \) [s] and \( E \) [J] are respectively the failure period and the energy due to Joule heating associated with the SEB. The error function \( \text{erf}(x) \) can be expanded in the following series.

\[
\text{erf}(x) = \frac{2}{\sqrt{\pi}} \sum_{n=0}^{\infty} \frac{(-1)^n \cdot x^{2n+1}}{n!(2n+1)}.
\]

This can be approximated as
\[
\text{erf}(x) \approx \frac{2}{\sqrt{\pi}} x \left( \text{if } x \leq \frac{\sqrt{\pi}}{2} \right) \\
\approx 1 \left( \text{if } x \geq \frac{\sqrt{\pi}}{2} \right)
\]

(8)

For a failure period of \(2.55 \times 10^{-9} \text{s}\), if \(a \geq \sqrt{4\pi D_{\text{Si}} t_f} \leq \frac{\sqrt{\pi}}{2}\), the size of the heat generation region is given by

\[
a \geq \sqrt{4\pi D_{\text{Si}} t_f} = 1.68 [\mu\text{m}].
\]

(9)

Therefore, when \(t_f \leq t_a = \frac{a^2}{4\pi D_{\text{Si}}}\), Eq. (5) can be simplified as

\[
T(t_f) = T_0 + \frac{1}{\rho C_p} \int_{t_0}^{t_f} \left( \frac{a}{\sqrt{4\pi D_{\text{Si}} \tau}} \right) \cdot V(\tau) d\tau
\]

\[
= T_0 + \frac{E}{\rho C_p \Delta t_f}.
\]

(10)

In contrast, when \(a \leq \sqrt{4\pi D_{\text{Si}} t_f} \leq \frac{\sqrt{\pi}}{2}\), the size of the heat generation region is

\[
a \leq \sqrt{4\pi D_{\text{Si}} t_f} = 1.68 [\mu\text{m}].
\]

(11)

The diameters of the damaged region in Fig. 7(b) and Fig. 8 are 1 \(\mu\text{m}\) and 0.34 \(\mu\text{m}\), respectively, both of which agree with Eq. (11).

In the case of \(t_a = t_b = \frac{a^2}{4\pi D_{\text{Si}}} \leq t_f \leq t_c\), Eq. (5) can be simplified using Eq. (8) as follows:

\[
T(t_f) = T_0 + \frac{E}{\rho C_p \Delta t_f} \int_{t_0}^{t_f} \left( \frac{a}{\sqrt{4\pi D_{\text{Si}} \tau}} \right) \cdot \text{erf} \left( \frac{b}{\sqrt{4\pi D_{\text{Si}} \tau}} \right) \cdot \text{erf} \left( \frac{c}{\sqrt{4\pi D_{\text{Si}} \tau}} \right) d\tau
\]

\[
= T_0 + \frac{E}{\rho C_p \Delta t_f} \left[ \int_{t_0}^{t_f} \left( \frac{a}{\sqrt{4\pi D_{\text{Si}} \tau}} \right) d\tau + \int_{t_f}^{t_c} \left( \frac{2}{\sqrt{\pi}} \frac{a}{\sqrt{4\pi D_{\text{Si}} \tau}} \right) d\tau \right]
\]

\[
= T_0 + \frac{E}{\rho C_p \Delta t_f} \left( 1 + \log \left( \frac{t_c}{t_f} \right) \right).
\]

(12)

The size of the damage region was then estimated using Eq. (12) and the simulated energy \(E\), which was calculated to be \(3.57 \times 10^7 \text{J}\) by integrating the product of the voltage and the SEB current shown in Fig. 3 from 0 to \(2.55 \times 10^{-9} \text{s}\). In addition, the potential distribution adopts a funnel-like shape during the SEB [19] [20]. The heat generation depth \(c\) is 150 \(\mu\text{m}\) in light of the potential deformation during the SEB shown in Fig. 9. Finally, the temperature \(T(t_f)\) reaches the melting point of Si (1683 K) after \(2.55 \times 10^{-9} \text{s}\), as shown in Fig. 3. By substituting \(E = 3.57 \times 10^7 \text{J}, t_f = 2.55 \times 10^{-9} \text{s}, T(t_f) = 1683 \text{K}\) and \(T_0 = 300 \text{K}\) into Eq. (12), we obtain an a value of 0.406 \(\mu\text{m}\), which is the same order of magnitude as the values of 1 \(\mu\text{m}\) and 0.34 \(\mu\text{m}\) in Fig. 7(b) and Fig. 8, respectively. The amount of initially generated charge due to recoil ions varies stochastically, and affects the size of the SEB damage region.

Fig. 10 shows the simulated SEB current, and the maximum electron and hole current density determined by TCAD simulations. The maximum current density for both electrons and holes is on the order of \(10^7 \text{A/cm}^2\). On the other hand, based on a maximum SEB current of 0.15 A and a Joule heating area of \(\pi a^2 = \pi (0.406 \mu\text{m})^2\), the maximum SEB current density is estimated to be \(2.9 \times 10^7 \text{A/cm}^2\), which is similar to the simulated current density shown in Fig. 10. Furthermore, the estimated current density \(2.9 \times 10^7 \text{A/cm}^2\) corresponds to the positive differential resistance (PDR) region shown in Fig. 11. These results support the validity of the proposed mechanism, by which SEB in power diodes occurs by thermal failure caused by local secondary breakdown.

Fig. 9. Electrostatic potential [V] at the times \(2.55 \times 10^{-9} \text{s}\) shown in Fig. 3.

Fig. 10. Simulated SEB current, and maximum electron and hole current density.
An analysis of neutron-induced SEB in Si power diodes has been carried out using a slice-and-view imaging technique. Unique annular microvoids were observed inside the devices, whose axially symmetric shape reflects spatially isotropic thermal diffusion. Analytical formulae for the local rise in temperature during SEB were derived from the thermal diffusion equation. The diameter of the damage region was theoretically estimated using these formulae together with the energy associated with Joule heating, which was calculated by TCAD device simulations. The estimated diameter of the damage region was similar to the observed sizes of the annular voids. Based on this diameter, the current density during SEB was calculated to be $2.9 \times 10^7 \text{ A/cm}^2$. This high current density corresponds to the positive differential resistance (PDR) region associated with secondary breakdown of the diodes. This holds for diode structures where the peak electric field can shift deep into the n$^+$ diffusion region, if this region has doping density which is of a not too steep gradient $dn/dx$. These results support the proposed mechanism by which SEB in power diodes occurs due to thermal failure caused by local secondary breakdown.

REFERENCES


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